

Influence of Pentavalent Niobium on Microstructural and Mechanical Properties of ZTA Ceramic Composite via Cold Isostatic Pressing

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ABSTRACT. The influence of pentavalent niobium (Nb₂O₅) on microstructural and mechanical properties of ZTA ceramic composite via cold isostatic pressing (CIP) method was investigated. The microstructural analysis and phase transformation was determined by scanning electron microscopy (SEM) and X-ray diffraction (XRD). XRD analysis revealed the presence of t-ZrO₂, m-ZrO₂ and Nb₂Zr₆O₁₇ phase. The maximum value of hardness, fracture toughness and flexural strength was found at 1 wt.% of Nb₂O₅ addition with the value of 1840 HV, 8.6 MPa.√m and 351 MPa, respectively. Thus, the addition of Nb₂O₅ results in an enhancement for the mechanical properties of ZTA ceramic.

Keywords: Hardness, Fracture toughness, Flexural strength, Cold isostatic pressing (CIP);

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1. INTRODUCTION

Zirconia toughened alumina (ZTA) is a zirconia-alumina system where the toughness of alumina is enhanced by adding zirconia [1]. Zirconia undergo transformation toughening which lead to an improvement on mechanical properties of the composites [2]. Pentavalent niobium (Nb₂O₅) is found as one potential additive that can improve the mechanical properties of ceramics [3-5]. Previous work done by Hassan et al. [3] found that Nb₂O₅ improved the mechanical properties of ZTA ceramic by sol gel method using uniaxial press as compaction method. For this work, cold isostatic pressing (CIP) method was chosen as compaction method in producing the homogeneity of powder compact [6,7]. CIP can improved the mechanical properties through lowering the porosity of ceramic compared to uniaxial press method [8-10]. Therefore, this study focuses on the role of CIP as compaction method on the microstructural and mechanical properties such as hardness, fracture toughness and flexural strength behaviour for ZTA added Nb₂O₅ ceramics.

2. MATERIALS AND METHODS

For this work, 80 wt.% of Al₂O₃ (Alcoa, A16SG, 99.0% purity) and 20 wt.% YSZ (Goodfellow, 94.6% purity) were used as starting materials. The amount of Nb₂O₅ (Merck, 99.5% purity) in weight percent was varied from 0 wt.% to 7 wt.% into the initial compositions by wet milling method. Then, the mixtures were dried to

100 °C and grinded to form powders. The mixed powders were compacted by uniaxial press at 15 MPa and followed by cold isostatic pressing (CIP) at 150 MPa for 2 min to form a cylindrical shapes with dimensions of 13 mm in diameter and 4 mm in height (for physical and microstructural characterization) and rectangular bars of dimensions of 40 × 10 × 10 mm (for mechanical evaluation). The green body was sintered in atmosphere at 1600 °C for 4 hours. The phase transformations of sintered body were analyzed by X-ray diffraction (XRD). Scanning electron microscopy (SEM) was used to observe the microstructure of each sample. The values of hardness and fracture toughness were measured by a Vickers indentation technique tester using a 30 kgf load. The fracture toughness values were calculated for all samples using the Palmqvist crack formula proposed by Niihara (1983)[11]:

$$3K_{Ic} = 0.035 (H^* a^{1/2}) (3E/H)^{0.4} (c/a)^{-0.5} \quad (1)$$

K_{Ic} is the fracture toughness, H is Vickers hardness, a is the half length of Vickers diagonal (μm), E is equal to the Young modulus of the samples and c is the length of the radial crack size (μm). Flexural strength was measured in a three-point bending test on a universal testing machine at a crosshead speed of 1 mm/min, and support distance of 20 mm.

3. RESULTS AND DISCUSSION

Table 1 shows the XRD analysis of the samples for different composition of Nb_2O_5 additions (0 - 7 wt.%). The phase existed in ZTA- Nb_2O_5 consist of alumina (Al_2O_3), tetragonal zirconia (t- ZrO_2), monoclinic zirconia (m- ZrO_2) and $\text{Nb}_2\text{Zr}_6\text{O}_{17}$ phase. Al_2O_3 are present as corundum matching ICSD pattern No. 98-005-7635 with hexagonal structure. Tetragonal zirconia and monoclinic zirconia was observed for all composition matching ICSD pattern No. 98-005-7635 and No. 98-007-1860. The XRD analysis also revealed the formation of secondary phase, $\text{Nb}_2\text{Zr}_6\text{O}_{17}$ (ICSD pattern No. 98-000-5068) starting at 3.0 wt.% Nb_2O_5 with orthorhombic structure. These identification phase's percentages correspond to the addition of Nb_2O_5 . The lower addition of Nb_2O_5 (0 - 1 wt.%) related to the effect of t- ZrO_2 and m- ZrO_2 phase, which is attributed to the toughening mechanism (t \rightarrow m). These mechanism are one of the factors to improve the toughness in ZTA- Nb_2O_5 composite [12]. Meanwhile, the higher Nb_2O_5 addition (3 - 7 wt.%) revealed the formation of $\text{Nb}_2\text{Zr}_6\text{O}_{17}$ phase. It was recorded that the highest formation of $\text{Nb}_2\text{Zr}_6\text{O}_{17}$ is 0.9 % at 7 wt.% Nb_2O_5 addition. This was proved by Energy dispersive X-ray analysis (EDX) in Fig. 2 with the presence of Nb element at YSZ grains.

Table 1 XRD analysis on phase composition for ZTA- Nb_2O_5

Nb_2O_5 addition (wt.%)	Phase Composition (%)			
	Al_2O_3	t- ZrO_2	m- ZrO_2	$\text{Nb}_2\text{Zr}_6\text{O}_{17}$
0	80.4	18.4	1.1	0
0.1	80.7	17.7	1.6	0
0.3	77.0	17.1	5.9	0
0.5	77.2	16.4	6.3	0
0.7	75.3	13.1	11.6	0
1	68.6	13.9	17.5	0
3	69.8	15.8	14.3	0.1
5	77.2	16.9	5.1	0.7
7	78.5	17.5	3.0	0.9

SEM micrographs for sintered samples of 0 wt.% and 7 wt.% of ZTA-Nb₂O₅ are shown in Fig. 1. Based on the microstructures, the grains for YSZ and Al₂O₃ are well distributed among each other. EDX analysis in Fig. 2 shows that the dark grains are Al₂O₃ and light grains are yttria stabilized zirconia (YSZ). The presence of Nb element was detected in YSZ grains, confirmed by the EDX analysis (Fig. 2). Meanwhile, Nb₂Zr₆O₁₇ phase found to be appeared at the YSZ grains when its amount exceeds the solubility limit.

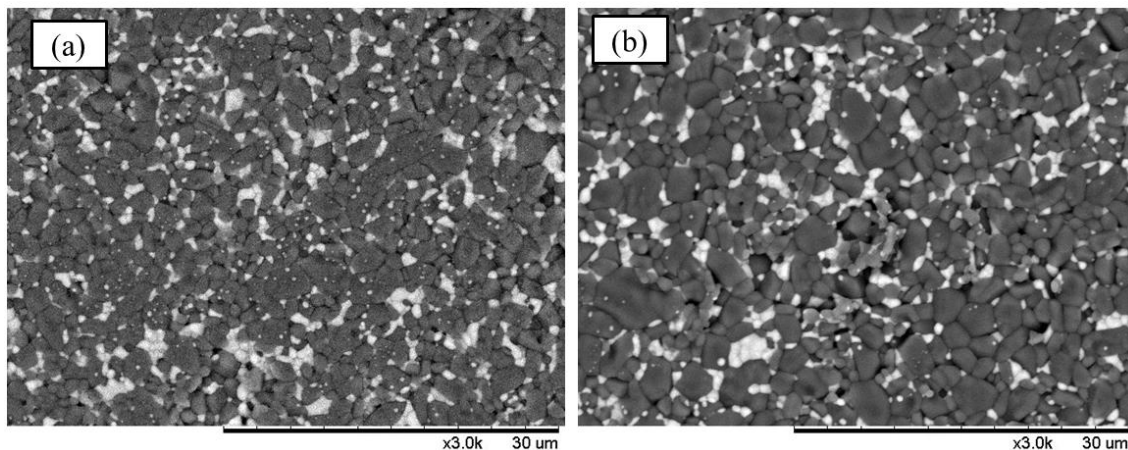


Fig. 1 SEM micrograph of ZTA- Nb₂O₅ composition (a) 0 wt.% and (b) 7.0 wt.%, at 3.0 K magnification

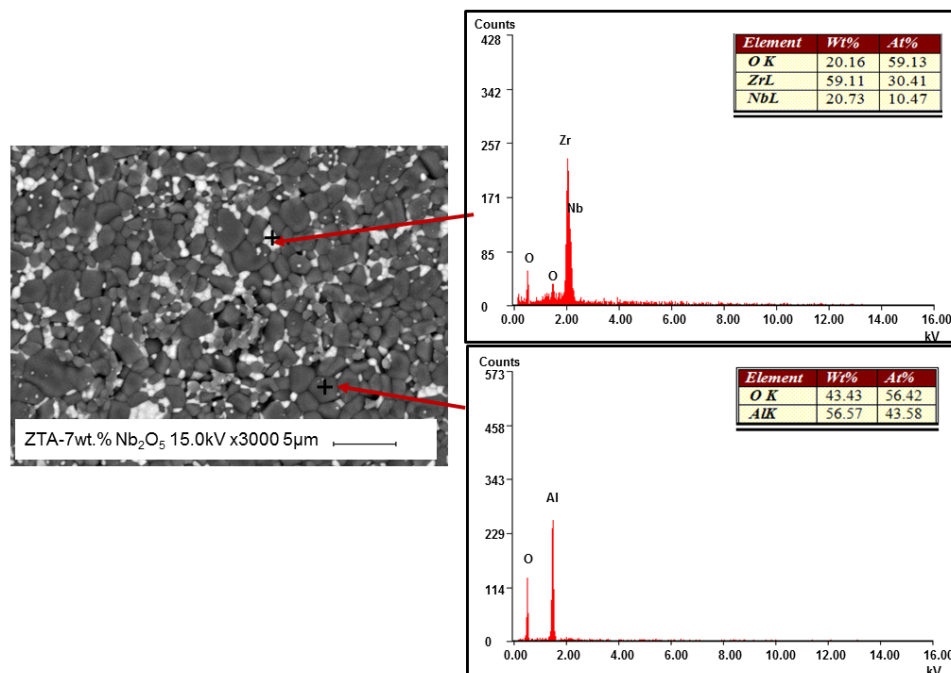


Fig. 2 Microstructure and EDX analysis for ZTA with 7 wt.% of Nb₂O₅ addition

Fig. 3 presents the results of hardness, fracture toughness and flexural strength of ZTA-Nb₂O₅ composites as a function of Nb₂O₅ content. As seen, the hardness, fracture toughness and flexural strength increased with increasing Nb₂O₅ content. The addition of 1 wt.% Nb₂O₅ gives the maximum value for hardness (1840 HV), fracture toughness (8.6 MPa.√m) and flexural strength (351 MPa). The dense compact through CIP enhanced the mechanical properties of this system, as the relative density achieved 97.4%-99.8% to the theoretical

density, where sample for 1 wt.% Nb_2O_5 gives the highest relative density (99.8%). CIP improved the particle packing and reorganized the microstructure, resulting in higher densification for sintered body [13]. Furthermore, the improvement of fracture toughness also can be related to the transformation toughening ($t \rightarrow m$) due to crack deflection and crack bridging that occurs at the crack path after indentation. However, the properties gradually decreased when reached at 3 wt.% of Nb_2O_5 addition. This phenomenon was related with the presence of secondary phase, $\text{Nb}_2\text{Zr}_6\text{O}_{17}$. A secondary phase can promote pore formation which might weaken the ceramics by reducing some of the mechanical properties. Pores were described as a discontinuity in the material, resulting in an undesirable effect [14]. This was confirmed by the results of porosity in Table 2, where the porosity found to increase at 3 wt.% of Nb_2O_5 addition.

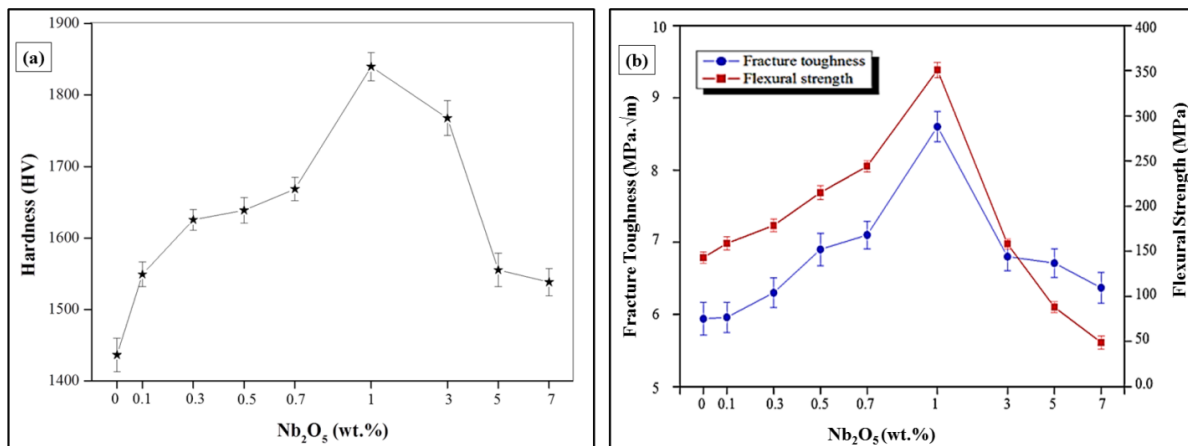


Fig. 3 Results of (a) hardness and (b) fracture toughness and flexural strength of ZTA-Nb₂O₅

Fig. 4 indicates the results of bulk density and theoretical density of ZTA samples with different addition of Nb_2O_5 . The result of bulk density follows the pattern of theoretical density. This result clearly indicates that Nb_2O_5 enhances sinterability and the density of ceramics. However, at 3 wt.% the densities decreased due to excess of Nb_2O_5 beyond the solubility limit, which results in the presence of $\text{Nb}_2\text{Zr}_6\text{O}_{17}$ phase (Table 1). The results show that by increasing the Nb_2O_5 addition, the bulk density increased and the percentage of porosity decreased as illustrated in Table 2. Lower percentage of porosity enhances the mechanical properties of the ZTA ceramic. The highest density (4.6 g/cm^3) with low percentage of porosity (0.08%), was found at 1 wt.% of Nb_2O_5 addition. Hassan et al. [3] stated that the ionic radius of Nb^{5+} (0.68 \AA) is smaller compared to Zr^{4+} (0.79 \AA) and Y^{3+} (1.019 \AA). When Nb^{5+} enter into the Zr^{4+} , it will cause a large amount of oxygen vacancies formed in the samples and accelerate the lattice diffusion, thus enhance the densification rate [15]. Furthermore, it also led to the formation of the observed new phase, confirmed by XRD result, which indicated in Table 1.

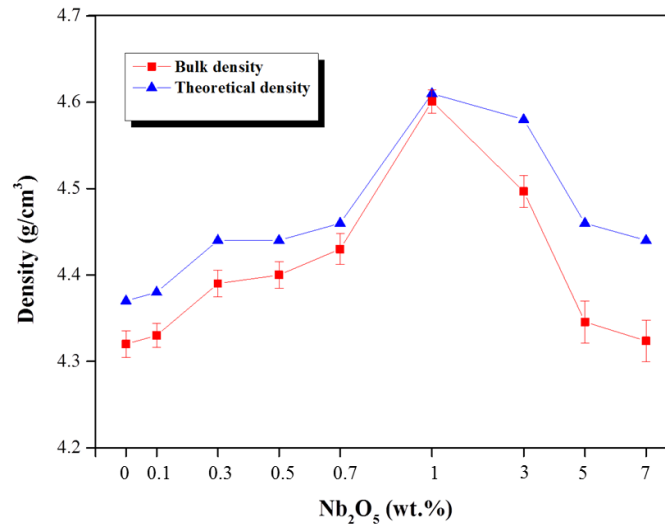


Fig. 4 Result of bulk density and theoretical density of ZTA-Nb₂O₅

Table 2 Result for porosity of ZTA-Nb₂O₅

Nb ₂ O ₅ addition (wt.%)	Porosity (%)
0	0.79
0.1	0.23
0.3	0.24
0.5	0.24
0.7	0.09
1	0.08
3	0.33
5	0.71
7	2.23

4. SUMMARY

The influence of pentavalent niobium (Nb₂O₅) on microstructural and mechanical properties of ZTA via CIP method was successfully investigated. The optimum composition are 1 wt.% Nb₂O₅ addition. The addition of Nb₂O₅ enhanced the hardness, fracture toughness and flexural strength compared to ZTA without Nb₂O₅ addition. The enhancements reach a limit at 1 wt.% addition, which is attribute to the transformation of t → m phase. These proved that the toughening mechanism occurs for 1 wt.% addition. However, the effectiveness of these toughening mechanism deteriorate the mechanical properties with further additions (3 - 7 wt.%) which is corresponds to an existence of Nb₂Zr₆O₁₇ phase.

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